## Rapid communication

# An investigation of the $\mathrm{Nd}_{2} \mathrm{O}_{3}-\mathrm{MoO}_{3}$ phase system: Thermal decomposition of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ and formation of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ 

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#### Abstract

A new neodymium molybdate, $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$, has been identified in the $\mathrm{Nd}_{2} \mathrm{O}_{3}-\mathrm{MoO}_{3}$ phase system. $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ appears to be a metastable phase, which does not form directly from a stoichiometric mixture of $\mathrm{Nd}_{2} \mathrm{O}_{3}$ and $\mathrm{MoO}_{3}$ oxides. Instead, it can be obtained by thermal decomposition of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15} . \mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ usually decomposes into $\mathrm{Nd}_{2}\left(\mathrm{MoO}_{4}\right)_{3}$, and the formation of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ critically depends on the heating regime used.

The structure of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ has been determined by single crystal X-ray diffraction. It crystallizes in the monoclinic space group $C 2 /$ $c$, with unit cell parameters of $a=12.425(1) \AA, b=19.860(2) \AA, c=13.882(1) \AA, \beta=100.767(2)^{\circ}, V=3365.2(5) \AA^{3}$ at 120 K . Nd atoms are seven and eight coordinate, and pairs of coordination polyhedra share edges and faces, respectively, to form $\mathrm{Nd}_{2} \mathrm{O}_{12}$ and $\mathrm{Nd}_{2} \mathrm{O}_{13}$ groups. All Mo atoms are in tetrahedral coordination environments, with some of the tetrahedra sharing corners to form pyromolybdate groups.


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## 1. Introduction

Rare-earth molybdates display a diverse and interesting chemistry. From a structural standpoint, these phases fall into two broad categories: conventional extended inorganic structures and high nuclearity molybdenum molecular clusters. A variety of interesting and exploitable properties are exhibited by rare-earth molybdates in both structural categories. Materials in the $R E_{2} \mathrm{MoO}_{6}$ family possess catalytic activity [1-3]. Compounds and solid solutions of the general formula $R E_{2}\left(\mathrm{MoO}_{4}\right)_{3}$ display negative thermal expansion [4], while the Gd and Tb phases are ferroelectric and ferroelastic $[5,6]$. The discovery of exceptional oxide ion conductivity in $\mathrm{La}_{2} \mathrm{Mo}_{2} \mathrm{O}_{9}$ has resulted in the preparation of numerous mixed rare-earth molybdates in this family [7]. Interesting electrical and magnetic behaviour can be found in molybdenum cluster compounds [8-10].

Two compounds having the general formula $R E_{6}$ $\mathrm{Mo}_{10} \mathrm{O}_{39}(R E=\mathrm{Ce}, \mathrm{Eu})$, with different, but related crystal

[^0]structures, have been reported in the literature [11-13]. This paper presents a new neodymium molybdate, $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$, the decomposition route to its formation and its structural characterization.

## 2. Experimental

Preheated $\mathrm{Nd}_{2} \mathrm{O}_{3}$ and $\mathrm{MoO}_{3}$ were mixed in a 1:4 stoichiometric ratio, thoroughly ground and fired overnight at $750^{\circ} \mathrm{C}$ in an open alumina crucible. A polycrystalline product, identified as $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ by powder X-ray diffraction, was formed. This powder was subsequently heated to $850^{\circ} \mathrm{C}$, slow cooled to $600^{\circ} \mathrm{C}$ at a rate of $3{ }^{\circ} \mathrm{C} / \mathrm{h}$ and then quenched to room temperature. Small pale purple crystals were isolated from the product. Their composition was found by single crystal X-ray diffraction to be $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$.

Solid state synthesis of polycrystalline $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ was attempted by mixing $\mathrm{Nd}_{2} \mathrm{O}_{3}$ and $\mathrm{MoO}_{3}$ in a $3: 10$ molar ratio, and heating the intimately ground mixture in an evacuated ampoule at $750^{\circ} \mathrm{C}$ for 4 days.

Progress of the solid state reactions and thermal decomposition of the product were monitored by powder X-ray diffraction. Room temperature data were collected on a Bruker AXS D8 Advance diffractometer with a Solex detector. Variable temperature powder X-ray diffraction experiments were performed using a Bruker AXS D8 Advance diffractometer equipped with a Vantec detector and an Anton Paar HTK1200 high temperature stage.

Single crystal X-ray diffraction data were collected on a Bruker AXS SMART diffractometer with an APEX CCD detector, equipped with a Bede Microsource ${ }^{\circledR}$ X-ray generator, using $\mathrm{Mo} K \alpha$ radiation. A crystal with approximate dimensions of $0.08 \times 0.08 \times 0.16 \mathrm{~mm}^{3}$ was selected for data collection. A full sphere of data was collected with a frame width of $0.3^{\circ}$ and a counting time of $20 \mathrm{~s} /$ frame. Data reduction was carried out using the SAINT [14] software suite. A multiscan absorption correction [15] was applied to the raw data and the resulting $R_{\text {int }}$ was $1.4 \%$. The crystal structure was solved by direct methods using

Table 1
Crystallographic details for $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$

| Temperature (K) | 120 |
| :--- | :--- |
| Crystal system | Monoclinic |
| Space group | $C 2 / c$ |
| $a(\AA)$ | $12.425(1)$ |
| $b(\AA)$ | $19.860(2)$ |
| $c(\AA)$ | $13.882(1)$ |
| $\beta(\mathrm{deg})$ | $100.767(2)$ |
| $V\left(\AA^{3}\right)$ | $3365.2(5)$ |
| $Z$ | 4 |
| Calc. $\rho\left(\mathrm{g} / \mathrm{cm}^{3}\right)$ | 4.833 |
| $\mu\left(\mathrm{~mm}^{-1}\right)$ | 12.774 |
| Total no. reflections | 22568 |
| No. unique reflections | 4928 |
| No. obs. reflections | 4253 |
| No. par. refined | 250 |
| $R_{\text {int }}(\%)$ | 1.4 |
| $R(\%)$ | 1.86 |
| wR $(\%)$ | 2.78 |

SIR92 [16] and refined in the crystals [17] software package. A total of 250 parameters (fractional coordinates, anisotropic atomic displacement parameters, an extinction parameter and scale) were refined to the final agreement factors of $R=1.86 \%, \mathrm{w} R=2.78 \%$. Full crystallographic details are given in Table 1.

## 3. Results and discussion

### 3.1. Decomposition of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ and formation of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$

Europium molybdate $\mathrm{Eu}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$, isostructural with the title phase, was isolated in the form of single crystals from a glassy melt obtained by heating a polyoxomolybdoeuropate precursor, $\left[\mathrm{Eu}_{2}\left(\mathrm{H}_{2} \mathrm{O}\right)_{12} \mathrm{Mo}_{8} \mathrm{O}_{27}\right] \cdot 6 \mathrm{H}_{2} \mathrm{O}$, at $800^{\circ} \mathrm{C}$ [13]. It was found that preparations were not very reproducible and the outcome depended on a number of factors, including the condition of the alumina container used, the exact composition of the starting material and the relative amounts of the precursor and seed crystals present. Attempts to prepare polycrystalline $\mathrm{Eu}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ from stoichiometric amounts of $\mathrm{Eu}_{2} \mathrm{O}_{3}$ and $\mathrm{MoO}_{3}$ by heating in evacuated quartz ampoules did not yield the desired product.

Similarly, our attempts to synthesize polycrystalline $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ by a solid state route from a stoichiometric ratio of $\mathrm{Nd}_{2} \mathrm{O}_{3}$ and $\mathrm{MoO}_{3}$ were not successful and $\mathrm{Nd}_{2}\left(\mathrm{MoO}_{4}\right)_{3}$ was obtained as the major phase instead.

As $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ single crystals were first isolated from heated $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$, a small amount of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ from the same batch was heated to near its melting at $950{ }^{\circ} \mathrm{C}$ and quenched. The product obtained formed a solid mass at the bottom of the alumina crucible. This material was removed from the crucible and ground, and shown by powder X-ray diffraction to be single phase $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$. Fig. 1 shows the Rietveld fit to the observed powder pattern, using the structural model obtained from single crystal diffraction.


Fig. 1. Rietveld plot for $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ : the observed data (blue), calculated pattern (red) and the difference curve (grey).

In this fitting procedure, only the unit cell parameters, instrument zero point, background terms, pseudo-Voigt profile function terms and an overall isotropic temperature factor were refined.

In order to understand the formation pathway of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ in more detail, we also investigated the behaviour of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ as a function of temperature by in situ powder X-ray diffraction. These measurements indicate that the decomposition of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ occurs gradually at temperatures above $500^{\circ} \mathrm{C}$. Fig. 2a shows the Rietveld profile of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ (with $\mathrm{Al}_{2} \mathrm{O}_{3}$ used as an internal standard), at room temperature, while Fig. 2b shows the refinement for the sample heated in situ to $600^{\circ} \mathrm{C}$.

The new phase in the X-ray diffraction pattern at $600^{\circ} \mathrm{C}$ can be identified as $\mathrm{Nd}_{2}\left(\mathrm{MoO}_{4}\right)_{3}$. Since no $\mathrm{MoO}_{3}$ reflections are present in the pattern and thermogravimetric analysis shows a gradual mass loss above $500^{\circ} \mathrm{C}$, the decomposition of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ presumably proceeds according to the equation:
$\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15} \rightarrow \mathrm{Nd}_{2}\left(\mathrm{MoO}_{4}\right)_{3}+\mathrm{MoO}_{3}$.

In the particular sample shown in Fig. 2b, about $26 \%$ of the starting molybdate has converted into $\mathrm{Nd}_{2}\left(\mathrm{MoO}_{4}\right)_{3}$ at $600^{\circ} \mathrm{C}$. Further in situ powder X-ray diffraction, with heating to $980^{\circ} \mathrm{C}$, did not reveal the formation of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$.
$\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ is a therefore a non-equilibrium phase which presumably occurs along the pathway:

$$
\begin{aligned}
3 \mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15} & \rightarrow \mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}+2 \mathrm{MoO}_{3} \\
& \rightarrow 3 \mathrm{Nd}_{2}\left(\mathrm{MoO}_{4}\right)_{3}+3 \mathrm{MoO}_{3}
\end{aligned}
$$

### 3.2. Description of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ structure

Fig. 3 shows two projection views and a perspective drawing of the unit cell of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$. Atomic coordinates and temperature factors are given in Table 2, and selected bond lengths in Table 3.

The structure of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ consists of seven and eight coordinate Nd atoms and $\mathrm{MoO}_{4}$ tetrahedra. Two unique Nd atoms are eight coordinate. $\mathrm{NdO}_{8}$ polyhedra share a face through three oxygen atoms, thus forming $\mathrm{Nd}_{2} \mathrm{O}_{13}$


Fig. 2. Rietveld profiles (observed, calculated and difference) for decomposition of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ : (a) $30{ }^{\circ} \mathrm{C}$ : $R_{\text {wp }}=10.071 \%$, the system contains $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ and $\mathrm{Al}_{2} \mathrm{O}_{3}$ internal standard, (b) $600{ }^{\circ} \mathrm{C}, R_{\mathrm{wp}}=10.111 \%$, the system contains $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}, \mathrm{Al}_{2} \mathrm{O}_{3}$ and $\mathrm{Nd}_{2}\left(\mathrm{MoO}_{4}\right)_{3}$.


Fig. 3. (a) Two projection views of the crystal structure of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ at 120 K : dark polyhedra represent tetrahedral $\mathrm{MoO}_{4}$ groups, light spheres are Nd atoms. (b) A perspective drawing: small circles at the corners of the dark tetrahedra represent O atoms. Positive directions of the three axes are indicated by axes labels.
units with a $\mathrm{Nd}(1)-\mathrm{Nd}(2)$ distance of $3.7755(4) \AA$ (Fig. 4a). The third unique Nd atom is seven coordinate and pairs of $\mathrm{Nd}(3) \mathrm{O}_{7}$ polyhedra share an edge, giving rise to $\mathrm{Nd}_{2} \mathrm{O}_{12}$ units with a $\mathrm{Nd}(3)-\mathrm{Nd}(3)^{\prime}$ distance of 3.7316 (4) $\AA$ (Fig. 4b).

All five independent Mo atoms are found in a tetrahedral coordination environment. Four of these are isolated, regular $\mathrm{MoO}_{4}$ groups, with $\mathrm{Mo}-\mathrm{O}$ bond lengths ranging from 1.73 to $1.81 \AA$, and $\mathrm{O}-\mathrm{Mo}-\mathrm{O}$ bond angles from $105^{\circ}$ to $115^{\circ}$. The remaining unique Mo atom is different, as its tetrahedra share a corner to form pyromolybdate
$\mathrm{Mo}(7)_{2} \mathrm{O}_{7}$ groups (Fig. 5), with three shorter $\mathrm{Mo}-\mathrm{O}$ distances of about $1.73 \AA$, a longer bond of $1.88 \AA$ to the shared oxygen atom and a $\mathrm{Mo}(7)-\mathrm{O}(14)-\mathrm{Mo}(7)$ bridging angle of $143^{\circ}$. The geometry of the pyromolybdate group is virtually identical to that found in $\mathrm{Eu}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$. Bond valence calculations [18] give sums of $3.19,3.35$ and 3.38 for the three unique Nd atoms, and 5.89, 5.87, 6.09, 6.00 and 5.96 for the five unique Mo atoms, indicating that, on average, the Nd-O bonds are under compressive stress, while the Mo-O bonds are under tensile stress. One

Table 2
Fractional atomic coordinates and atomic displacement parameter isotropic equivalents for $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$

| Atom | $x$ | $y$ | $z$ | $U_{\text {eq }}\left(\AA^{2}\right)$ |
| :--- | :--- | :--- | :--- | :--- |
| Nd1 | $0.62608(1)$ | $0.155268(9)$ | $0.64678(1)$ | 0.0047 |
| Nd2 | $0.41289(1)$ | $0.148771(9)$ | $0.41431(1)$ | 0.0048 |
| Nd3 | $0.91044(1)$ | $0.058904(9)$ | $0.41602(1)$ | 0.0057 |
| Mo4 | $0.10326(2)$ | $0.21643(1)$ | $0.36184(2)$ | 0.0052 |
| Mo5 | $0.59452(2)$ | $-0.02014(1)$ | $0.36135(2)$ | 0.0050 |
| Mo6 | $0.18431(2)$ | $-0.00059(1)$ | $0.36069(2)$ | 0.0055 |
| Mo7 | $0.91358(2)$ | $0.15382(1)$ | $0.83909(2)$ | 0.0070 |
| Mo8 | $0.69320(2)$ | $0.19286(1)$ | $0.38136(2)$ | 0.0063 |
| O9 | $0.6072(2)$ | $0.1706(1)$ | $0.4682(2)$ | 0.0077 |
| O10 | $0.6087(2)$ | $0.2516(1)$ | $0.7426(2)$ | 0.0093 |
| O11 | $0.6040(2)$ | $0.0543(1)$ | $0.74340(2)$ | 0.0096 |
| O12 | $0.2390(2)$ | $0.2016(1)$ | $0.4192(2)$ | 0.0100 |
| O13 | $0.1260(2)$ | $0.0025(1)$ | $0.2350(2)$ | 0.0118 |
| O14 | 1.00 | $0.1235(2)$ | 0.75 | 0.0105 |
| O15 | $0.7285(2)$ | $0.0048(1)$ | $0.4146(2)$ | 0.0103 |
| O16 | $1.0783(2)$ | $-0.0022(1)$ | $0.9337(2)$ | 0.0095 |
| O17 | $0.2781(2)$ | $0.0646(1)$ | $0.3863(2)$ | 0.0122 |
| O18 | $0.9652(2)$ | $0.1383(1)$ | $1.1458(2)$ | 0.0089 |
| O19 | $0.7734(2)$ | $0.1456(1)$ | $0.7992(2)$ | 0.0103 |
| O20 | $0.7521(2)$ | $0.0776(1)$ | $0.6077(2)$ | 0.0103 |
| O21 | $0.7105(2)$ | $0.2800(1)$ | $0.3740(2)$ | 0.0107 |
| O22 | $0.4571(2)$ | $0.0842(1)$ | $0.5658(2)$ | 0.0084 |
| O23 | $0.4549(2)$ | $0.22095(1)$ | $0.5699(2)$ | 0.0094 |
| O24 | $0.9456(2)$ | $0.2379(1)$ | $0.8637(2)$ | 0.0183 |
| O25 | $0.5158(2)$ | $0.0528(1)$ | $0.3594(2)$ | 0.0081 |
| O26 | $0.9534(2)$ | $0.1080(2)$ | $0.9458(2)$ | 0.0168 |
| O27 | $0.6396(3)$ | $0.1593(2)$ | $0.2664(2)$ | 0.0204 |
| O28 | $0.8240(2)$ | $0.1590(1)$ | $0.4234(2)$ | 0.0164 |
|  |  |  |  |  |
|  |  |  |  |  |

numerical indicator of the discrepancies between the bond valence sums found in a structure and the expected values is the global instability index, GII. Values of GII smaller than $0.05 \mathrm{v} . \mathrm{u}$. correspond to structures with virtually no lattice-induced strain, while values of $0.20 \mathrm{v} . \mathrm{u}$. and above suggests that the strain is sufficiently large for the structure to be unstable [19]. The calculated value of GII for $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ is 0.15 and this might be related to the apparent metastable character of this phase. Similar behaviour is found in the $R E_{2} \mathrm{MoO}_{6}$ family of rare-earth molybdates [20].
$\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ is isostructural with the analogous Eu compound, while $\mathrm{Ce}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ has a slightly different, although closely related structure in space group $P-1$ [12,13]. The local coordination of Mo atoms in both structure types is the same: most of the tetrahedral $\mathrm{MoO}_{4}$ groups are isolated, with one pair forming $\mathrm{Mo}_{2} \mathrm{O}_{7}$ groups. On the other hand, environments of the rare-earth cations differ. In $\mathrm{Ce}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$, there are six unique Ce atoms and they are all eight-coordinate. Two unique pairs of $\mathrm{CeO}_{8}$ polyhedra share a face via three O atoms, forming $\mathrm{Ce}_{2} \mathrm{O}_{13}$ units similar to those found in the title compound and the Eu analogue. A different $\mathrm{CeO}_{8}$ pair share a face through four O atoms, giving rise to $\mathrm{Ce}_{2} \mathrm{O}_{12}$ groups with a short $\mathrm{Ce}-\mathrm{Ce}$ separation of $3.69 \AA$. The remaining $\mathrm{CeO}_{8}$ pair share an edge, with the two central Ce atoms $4.16 \AA$ apart.

Table 3
Selected bond lengths and bond valence sums for $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$

| M | O | $d(\AA)$ | BVS (v.u.) |
| :---: | :---: | :---: | :---: |
| Nd1 | O21 | 2.465(3) | 3.19 |
|  | O9 | $2.465(2)$ |  |
|  | O10 | 2.362(2) |  |
|  | O11 | 2.461(3) |  |
|  | O19 | 2.533(2) |  |
|  | O20 | 2.333(3) |  |
|  | O22 | $2.605(2)$ |  |
|  | O23 | 2.553(3) |  |
| Nd2 | O27 | 2.480(3) | 3.35 |
|  | O24 | 2.414(3) |  |
|  | O9 | 2.429(2) |  |
|  | O12 | 2.414(3) |  |
|  | O17 | 2.346 (3) |  |
|  | O22 | 2.437(2) |  |
|  | O23 | 2.564(2) |  |
|  | O25 | 2.492(2) |  |
| Nd3 | O26 | 2.506(3) | 3.38 |
|  | O18 | 2.471(2) |  |
|  | O16 | 2.395(3) |  |
|  | O13 | 2.346 (3) |  |
|  | O16 | 2.342(2) |  |
|  | O15 | 2.499(3) |  |
|  | O28 | 2.272(3) |  |
| Mo4 | O23 | $1.795(2)$ | 5.89 |
|  | O10 | 1.787(2) |  |
|  | O18 | 1.763(2) |  |
|  | O12 | $1.750(3)$ |  |
| Mo5 | O22 | $1.815(2)$ | 5.87 |
|  | O11 | $1.789(2)$ |  |
|  | O15 | 1.762(3) |  |
|  | O25 | $1.745(2)$ |  |
| Mo6 | O16 | 1.807(3) | 6.09 |
|  | O20 | 1.740 (3) |  |
|  | O13 | $1.761(3)$ |  |
|  | O17 | $1.735(3)$ |  |
| Mo7 | O14 | 1.882(1) | 5.96 |
|  | O19 | 1.734(3) |  |
|  | O24 | $1.737(3)$ |  |
|  | O26 | $1.730(3)$ |  |
| Mo8 | O9 | 1.809(2) | 6.00 |
|  | O21 | 1.749 (3) |  |
|  | O27 | 1.744 (3) |  |
|  | O28 | 1.754(3) |  |

The existence of different, but related crystal structure types for $R E_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ compounds is presumably related to the preference of the larger rare-earth cations such as $\mathrm{Ce}^{3+}$ for a higher average coordination number relative to the smaller $R E^{3+}$ species. Similar instances of polymorphism in rare-earth molybdates can be found in the $R E_{2} \mathrm{MoO}_{6}$ and $R E_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ structural families [20,21].

## 4. Conclusions

An investigation of the $\mathrm{Nd}_{2} \mathrm{O}_{3}-\mathrm{MoO}_{3}$ phase system has led to the isolation of a new neodymium molybdate,


Fig. 4. Coordination environment of Nd atoms in $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ : (a) $\mathrm{Nd}_{2} \mathrm{O}_{13}$ groups formed by face-sharing of two $\mathrm{NdO}_{8}$ polyhedra; (b) $\mathrm{Nd}_{2} \mathrm{O}_{12}$ groups formed by edge-sharing of two $\mathrm{NdO}_{7}$ polyhedra.


Fig. 5. A pyromolybdate group in $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$.
$\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$, both in the form of single crystals and as a polycrystalline product. Variable temperature powder X -ray diffraction studies have shown that $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ is a non-equilibrium phase, which can form on the pathway
of decomposition of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ into $\mathrm{Nd}_{2}\left(\mathrm{MoO}_{4}\right)_{3}$. The successful isolation of the title phase depends critically on the precise heating regime used, as experiments with a number of different samples of $\mathrm{Nd}_{2} \mathrm{Mo}_{4} \mathrm{O}_{15}$ have shown that it is possible to decompose it into $\mathrm{Nd}_{2}\left(\mathrm{MoO}_{4}\right)_{3}$ and continue slow-heating to the melting point without the formation of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$.

The structure of $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ has been determined by single crystal X-ray diffraction. It contains seven- and eight-coordinate Nd atoms, whose coordination polyhedra share edges and faces to form $\mathrm{Nd}_{2} \mathrm{O}_{12}$ and $\mathrm{Nd}_{2} \mathrm{O}_{13}$ groups. All Mo atoms are tetrahedral, with some of the $\mathrm{MoO}_{4}$ tetrahedra sharing vertices to form $\mathrm{Mo}_{2} \mathrm{O}_{7}$ groups. $\mathrm{Nd}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$ is isostructural with $\mathrm{Eu}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$, and closely related to $\mathrm{Ce}_{6} \mathrm{Mo}_{10} \mathrm{O}_{39}$.

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